## Supporting information

The Interplay between (Electro)chemical and (Chemo)mechanical Effects in the Cycling Performance of Thiophosphate-based Solid-State Batteries

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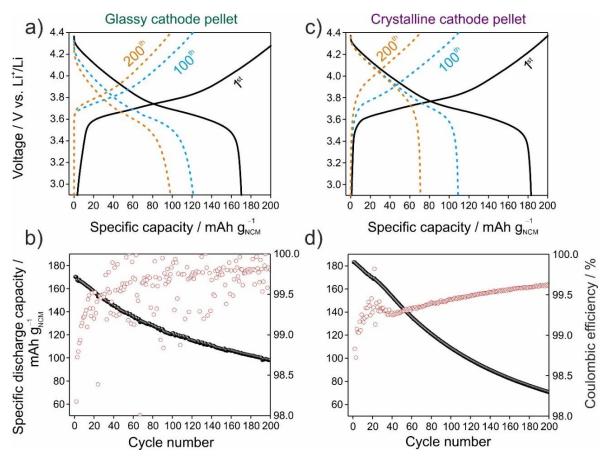
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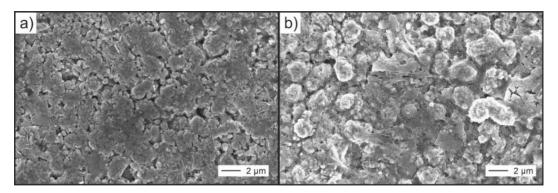
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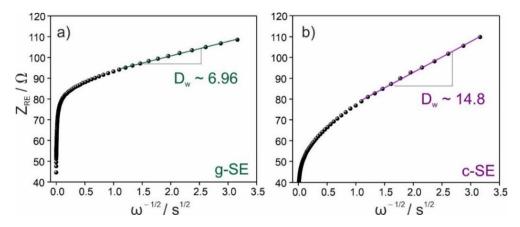
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**Figure S1.** Representative 1<sup>st</sup>, 100<sup>th</sup>, and 200<sup>th</sup> cycle charge/discharge curves at a rate of C/5 and 45 °C of pelletized SSB cells with (a) glassy SE (1.5Li<sub>2</sub>S-0.5P<sub>2</sub>S<sub>5</sub>-LiI) and (c) crystalline SE (Li<sub>6</sub>PS<sub>5</sub>CI) and (b, d) corresponding specific discharge capacities and Coulombic efficiencies over 200 cycles.



**Figure S2.** Top-view SEM images of the cathode of SSB cells in (a) pelletized and (b) slurry-cast setups.



**Figure S3.** The real part of impedance data plotted versus the  $(frequency)^{-1/2}$  and corresponding linear fit at low frequencies for (a) the g-SE cell and (b) the c-SE cell after 200 cycles at a rate of C/5 and 45 °C.

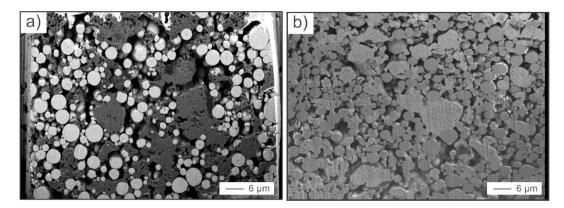
The following equations correlate the Warburg coefficient to the contact area at the CAM/SE interface:<sup>[1-3]</sup>

$$Z_{\text{RE}}(\omega) = R_{\text{SE}} + R_{\text{CT}} + D_{\text{W}} \cdot \frac{1}{\sqrt{\omega}}$$
 (Eq. S1),

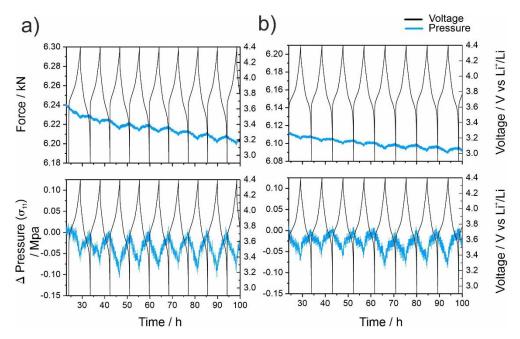
with  $Z_{\text{RE}}(\omega)$  being the real part of the impedance,  $R_{\text{SE}}$  the SE bulk resistance,  $R_{\text{CT}}$  the charge-transfer resistance,  $D_{\text{W}}$  the Warburg coefficient, and  $\omega$  the frequency.

$$D_{\rm w} = \frac{RT}{n^2 F^2 A \sqrt{2}} \left( \frac{1}{c_{\rm i} \sqrt{D_{\rm i}}} \right) \tag{Eq. S2}$$

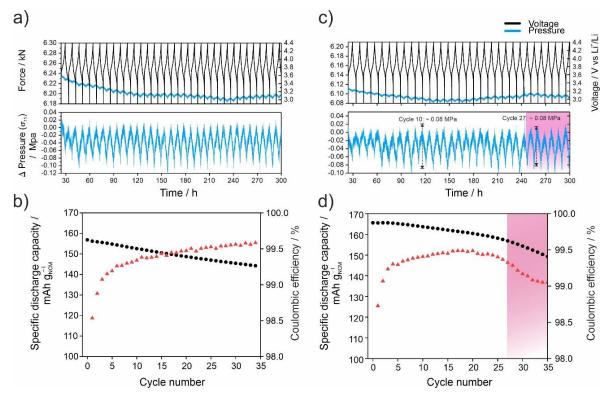
*R* is the gas constant, *T* the absolute temperature, *n* the number of electrons exchanged in the redox process, *F* the Faraday constant, *A* the contact area,  $D_i$  the lithium-diffusion coefficient in the bulk electrode material, and  $c_i$  represents the concentration of lithium ions in the bulk electrode material.



**Figure S4.** Cross-sectional FIB-SEM slice images of slurry-cast cathodes with (a) glassy SE (1.5Li<sub>2</sub>S-0.5P<sub>2</sub>S<sub>5</sub>-LiI) and (b) crystalline SE (Li<sub>6</sub>PS<sub>5</sub>CI). Note that the cathode was not cold-pressed prior to the measurement. More pores within the SE particles are observed for the c-SE cell than the g-SE cell. Panel (a) shows a backscattered electron image and panel (b) is a secondary electron image.



**Figure S5.** Raw data of the uniaxial force of slurry-cast cathodes with (a) glassy SE (1.5Li<sub>2</sub>S-0.5P<sub>2</sub>S<sub>5</sub>-Lil) and (b) crystalline SE (Li<sub>6</sub>PS<sub>5</sub>Cl) recorded during cycling and the corresponding change in uniaxial stress ( $\sigma_{11}$ ) after baseline correction. SSB cells tested at 45 °C, C/5, 2.9-4.4 V vs Li<sup>+</sup>/Li.



**Figure S6.** Force and pressure response during cycling of slurry-cast cathodes with (a) glassy SE ( $1.5Li_2S-0.5P_2S_5-LiI$ ) and (c) crystalline SE ( $Li_6PS_5CI$ ) and (b, d) corresponding specific discharge capacities and Coulombic efficiencies over 35 cycles. SSB cells tested at 45 °C, C/5, 2.9-4.4 V vs Li<sup>+</sup>/Li.

Electrochemical decomposition of Li<sub>2</sub>CO<sub>3</sub> impurities:

 $2\text{Li}_2\text{CO}_3 \rightarrow 4\text{Li}^+ + 4\text{e}^- + 2\text{CO}_2 \uparrow + \text{O}_2 \uparrow \qquad (\text{Eq. S3})$ 

## References

[1] Wang S, Zhang W, Chen X, Das D, Reuss R, Gautam A, Walther F, Ohno S, Koerver R, Zhang Q, Zeier W G, F H Richter, Nan C-W and Janek J. 2021 *Adv. Energy Mater.* **11** 2100654.

[2] Wang P-P, Xu C-Y, Li W-D, Wang L and Zhen L. 2015 *Electrochim. Acta* **169** 440-446.

[3] Xiao P, Lv T, Chen X and Chang C. 2017 Sci. Rep. 7 1408.